

Graphene Epoxy-Based Composites as Efficient Electromagnetic Absorbers in the Extremely High-Frequency Band

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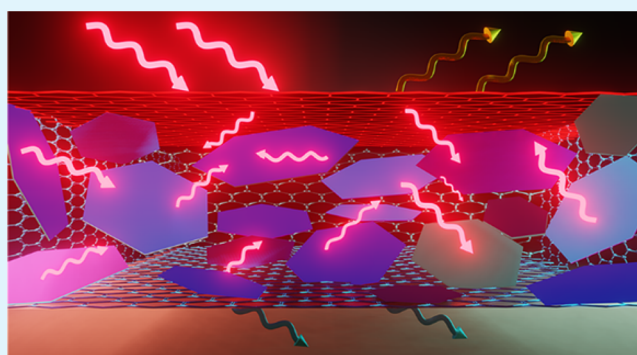
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Supporting Information

ABSTRACT: We report on the synthesis of the epoxy-based composites with graphene fillers and test their electromagnetic shielding efficiency by the quasi-optic free-space method in the extremely high-frequency (EHF) band (220–325 GHz). The curing adhesive composites were produced by a scalable technique with a mixture of single-layer and few-layer graphene layers of few-micrometer lateral dimensions. It was found that the electromagnetic transmission, T , is low even at small concentrations of graphene fillers: $T < 1\%$ at a frequency of 300 GHz for a composite with only $\phi = 1$ wt% graphene. The main shielding mechanism in composites with the low graphene loading is absorption. The composites of 1 mm in thickness and a graphene loading of 8 wt% provide an excellent electromagnetic shielding of 70 dB in the sub-terahertz EHF frequency band with negligible energy reflection to the environment. The developed lightweight adhesive composites with graphene fillers can be used as electromagnetic absorbers in the high-frequency microwave radio relays, microwave remote sensors, millimeter wave scanners, and wireless local area networks.

KEYWORDS: graphene, electromagnetic shielding, electromagnetic absorbers, extremely high-frequency band, sub-THz frequency, electrical percolation



I. INTRODUCTION

Rapid development of the wireless communications, distributed sensor arrays, and portable electronic devices made the control of electromagnetic (EM) radiation and reduction of EM pollution crucially important.^{1–3} The electromagnetic interference (EMI) shielding is needed to ensure that electronics operate reliably and without detrimental effects on human health.^{4–8} Different frequency ranges and types of devices require different solutions for EMI shielding. The shielding of EM energy at high-frequency bands can be particularly challenging. In many cases, additional requirements are imposed on the materials used for EMI shielding, including thickness and weight limits, mechanical and thermal properties, and electrical conduction or isolation. Absorption of EM energy rather than its reflection back to the environment offers benefits in a wide range of commercial and defense applications. However, many existing EMI shielding materials, e.g., metallic coatings, simply redirect the EM energy via the electrical conduction-based reflection mechanism. The latter shifts the problem of EM pollution from one element to another, and while protecting electronic components, it can negatively affect the human health.^{4–8} The metal-based EMI shielding materials have other problems

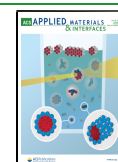
associated with heavy weight, corrosion, and difficulty of processing.

An alternative to metal shielding approach is the use of polymer-based materials with electrically conductive fillers.^{9–16} Such composite materials demonstrated effective EMI shielding in MHz and lower GHz ranges. The initial research was conducted on composites with metal particles, which were added as fillers in high weight fractions to increase the electrical conductivity.^{10,13,16–19} However, the polymer–metal composites suffer from relatively high weight, cost, and corrosion. More recent studies reported the use of carbon fibers,^{20–29} carbon black,^{28,30} bulk graphite,^{31–33} carbon nanotubes,^{34–40} reduced graphene oxide,^{41–54} graphene,^{55–58} and combination of carbon allotropes with or without other metallic or nonmetallic particles.^{43–45,55,57,59–65} A new class of quasi-two-dimensional materials, MXenes, have also been shown to exhibit high EM shielding efficiencies when added

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as fillers to polymer matrices or deployed as thin membranes.¹⁷ However, the dominant mechanism of EMI shielding in these materials is reflection, which make them less attractive for a wide range of applications where EM reflection creates problems. The studies of graphene-enhanced composites for EMI shielding mostly focused on a super high-frequency (SHF) range, specifically X-band.^{53–58,66,67} The reported studies covered the range of frequencies from MHz to 30 GHz. We are aware of one study of the use of graphene composites in the sub-THz range.⁶⁸ One can conclude from the reported studies that effectiveness and physical mechanisms of shielding differ substantially depending on the EM frequency range, graphene loading fraction, and characteristics such as size and thickness of the fillers. In our prior work, we demonstrated a dual function of graphene composites for the X-band, which includes EMI shielding and heat removal.³ We are not aware of the reports that would address the EM wave interaction with the polymer-based graphene composites in the sub-terahertz extremely high-frequency (EHF) band.

Here, we report the results of the investigation of the EMI shielding efficiency of graphene epoxy-based composites in the EHF band with the frequencies from 220 to 320 GHz, which correspond to the WR-3 band in microwave waveguide classification. This sub-terahertz frequency band is important for radio astronomy, high-frequency microwave radio relay, microwave remote sensing, millimeter wave scanning, and wireless local area networks. From the physics point of view, this frequency band is interesting because of possible changes in the electrical percolation in composites as compared to percolation in DC or low-frequency regime. We found that the synthesized graphene-enhanced composites are not only efficient EMI shielding materials but also achieve their function by absorbing EMI waves rather than by reflecting them. Moreover, it appears that there exists an optimum loading fraction of graphene, near 1 wt%, at which the material effectively absorbs but not reflects the EM waves. The examined epoxy-based curing composites can serve as adhesives for the electronic components while performing the EMI shielding functions.

II. MATERIAL SYNTHESIS

In the context of the present study, we use the term graphene for a mixture of single-layer graphene (SLG) and few-layer graphene (FLG). Commercially available FLG flakes (xGnP H-25, XG Sciences, US) with an average surface area of $\sim 65 \text{ m}^2\text{g}^{-1}$ and average lateral dimensions of $25 \mu\text{m}$ were added to the base epoxy resin (bisphenol A-(epichlorhydrin); molecular weight, ≤ 700 ; Allied High Tech Products, Inc.) at precalculated proportions for each composite sample with the specific FLG loading. The compounds were mixed using a high-shear speed mixer (FlackTek Inc.) at 800 rpm for 3 min and vacuumed for 10 min to extract the trapped air bubbles. This procedure was repeated three times to obtain a void-free mixture. Then, the curing agent (triethylenetetramine, Allied High Tech Products, Inc.) was added in a mass ratio of 12:100 with respect to the epoxy resin. The compound was mixed and vacuumed one more time and left in the oven for $\sim 2 \text{ h}$ at 70°C to cure and solidify. More detailed analysis of the size and thickness distribution of SLG and FLG in a mixture has been reported by some of us earlier in the context of thermal studies^{69–72} and can be found in the Supporting Information. The final composite samples were all in the form of disks with a diameter of 25.6 mm and thicknesses of $\sim 1 \text{ mm}$. The sample

thickness affects the total absorption and the total shielding efficiency of the composites. The schematic of the shielding function of the materials, optical images of the samples, and Raman spectroscopy data are presented in Figure 1. The

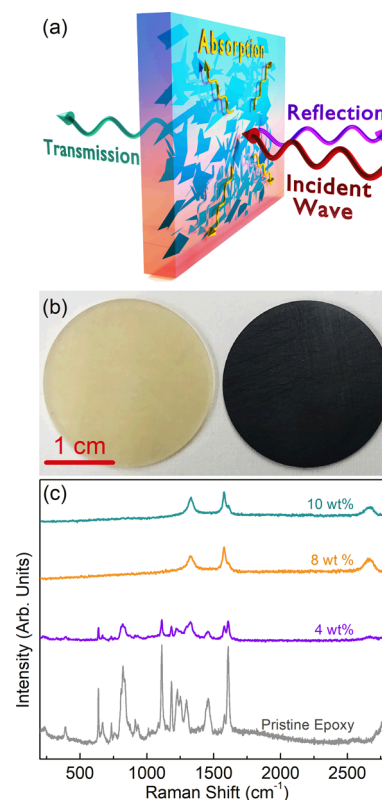


Figure 1. (a) Schematic of the interaction of the EM waves with the epoxy-based composites with the low filler loading of graphene. (b) Optical image of the pristine epoxy (left) and epoxy with $f = 4 \text{ wt}\%$ graphene (right). (c) Raman spectra of epoxy with different loading fractions of graphene fillers. Note the color change of the epoxy composite with the addition of graphene. The composites can be used as adhesives for mounting microwave devices and as a coating layer.

Raman spectrum of representative composite samples confirms the composition via the characteristic peaks of the epoxy and graphene as well as the expected change in concentration evidenced from the intensity increase of the G peak and 2D band of graphene in the samples with the higher graphene loading. Additional material characterization data for the graphene-enhanced composites, including representative scanning electron microscopy (SEM) images are provided in the Supporting Information.

III. RESULTS OF THE EM MEASUREMENTS

The shielding efficiency of the epoxy-based graphene composites was determined from the scattering parameters measured using the quasi-optic free-space method.^{73–75} The measurement setup consisted of a vector network analyzer with a pair of frequency extenders, two high gain horn antennas and two double convex lenses. The measurements were performed in the range of frequencies from 220 to 320 GHz. To obtain the reflection, R , and transmission, T , coefficients, three measurements were performed. The first measurement was conducted with the sample, the second one was without the sample, and the third measurement was with a reference plane

metal reflector. Two last measurements were used as the references to calculate the transmission and reflection coefficients, respectively. The reference measurements allow one to compensate for the transmission losses in the measurement path. The transmission and reflection coefficients are calculated according to the standard equations³⁸

$$T = \frac{|S_{21s}|^2}{|S_{21e}|^2} \quad (1)$$

$$R = \frac{|S_{11s}|^2}{|S_{11m}|^2} \quad (2)$$

where S_{11s} and S_{21s} are the results for the measurements with the sample, S_{21e} is the result for the measurement with an empty optical path, and S_{11m} is the result for the measurement with a metal plate.

In the case of multiple reflections in the quasi-optical path, the measurement data can be affected. To account for this possibility, an additional data processing step was applied. It consisted of the time domain gating.⁷⁶ The latter was possible owing to the broad frequency range and a large number of the measurement points (up to 32,000). The measured complex scattering parameters were transformed to the time domain. After that, an appropriate time domain window was applied. Finally, the time-gated data were transformed back to the frequency domain. This approach allowed us to improve the accuracy and reliability of the obtained data for the transmission and reflection coefficients. The obtained R and T coefficients were used for the calculation of the absorption coefficient, A , and the effective absorption coefficient, A_{eff} , which are given as³⁸

$$A = 1 - R - T \quad (3)$$

$$A_{\text{eff}} = \frac{A}{1 - R} = \frac{1 - R - T}{1 - R} = \frac{A}{A + T} \quad (4)$$

Both parameters define the EM absorption characteristics of the shielding material. One should note that A_{eff} describes the actual absorption properties of the material since part of the EM energy is reflected from the surface of the material.

Figure 2a,b shows the reflection, R , and transmission, T , coefficients for pristine epoxy and epoxy-based composites with the graphene loading ranging from 0.8 to 8.0 wt%. Figure 2c shows the lower bound of the transmission coefficient ($T < 5\%$), specifically allowing to distinguish the transmission coefficients for the composites with the graphene loading above 1 wt%. One can see that a small addition of graphene (0.8–1.0 wt%) does not change noticeably the EM wave reflection from the samples but does significantly reduce the transmission. The transmission coefficient decreases from 60% for the pristine epoxy to less than 5% for the epoxy-based composite with only 1 wt% graphene fillers (see both Figure 2b,c). The transmission monotonically decreases with the frequency for the composites with a graphene loading of 1 wt% or more in the considered frequency band.

Figure 3a,b shows the absorption coefficient, A , and the effective absorption coefficient, A_{eff} , respectively. Both parameters define the EM absorption characteristics of the shielding material. Figure 3c shows the upper bound of the effective absorption coefficient ($A_{\text{eff}} > 95\%$), specifically allowing to distinguish the values of A_{eff} for the composites with the graphene loading above 1 wt%. The data in Figure 3 make it

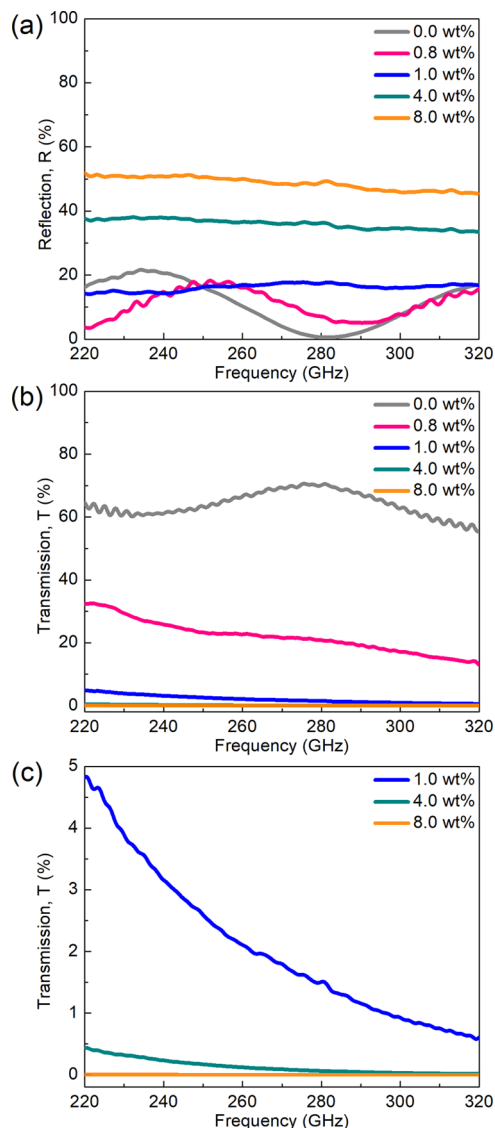


Figure 2. Coefficients of (a) reflection and (b) transmission for composites with different graphene loadings in the 220–320 GHz frequency range. (c) Lower bound of the transmission coefficient ($T < 5\%$), allowing to distinguish the transmission coefficients for the composites with the graphene loading above 1 wt%.

clear that the main EM wave shielding mechanism in composites with the low graphene loading (0.8–1.0 wt%) is absorption. The absorption coefficient, A , increases from 20% for the pristine epoxy to more than 80% for the epoxy-based composite with only 1 wt% graphene fillers (see Figure 3a). The absorption monotonically increases with the frequency for the composites with a graphene loading of 1 wt% or more in this frequency band (see both Figure 3b,c). The data presented in Figures 2 and 3 indicate that composites with the low loading of graphene provide efficient EMI shielding in this frequency band via absorption with small energy reflection to the environment. A graphene loading of 1 wt% appears to be the optimum for the tested composites with the given size and thickness of the fillers.

Figure 4a,b presents the shielding efficiency of the composites by reflection, SE_R , and absorption, SE_A , respectively. The latter includes the internal reflections of the EM waves inside the composite medium. The total shielding

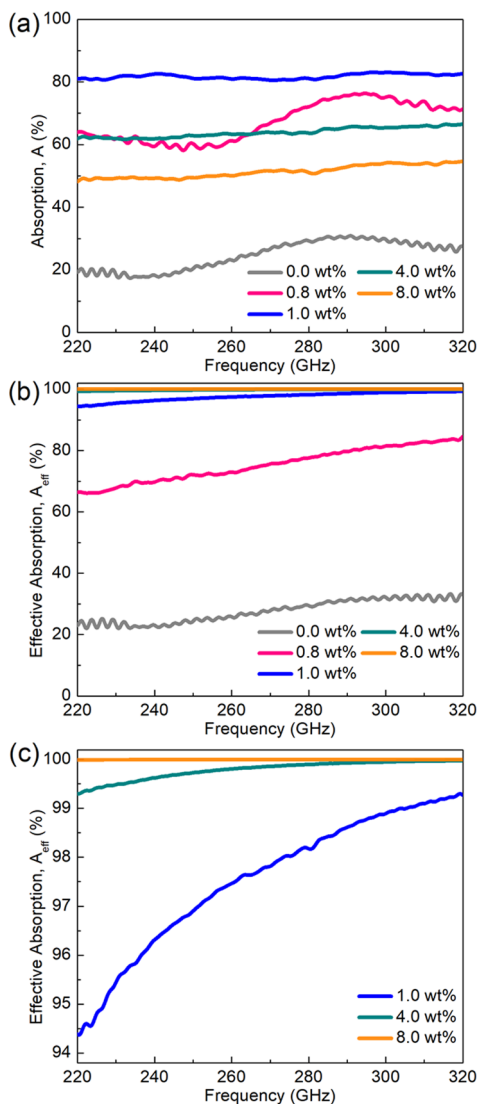


Figure 3. Coefficients of (a) absorption and (b) effective absorption for composites with different graphene loading fractions. (c) Upper bound of the effective absorption coefficient ($A_{\text{eff}} > 95\%$), allowing to distinguish the values of A_{eff} for the composites with the graphene loading above 1 wt%.

efficiency, SE_T , is the sum of shielding by reflection and absorption, and it is plotted as a function of frequency for different filler loadings in Figure 4c. For these plots, the shielding efficiency parameters were calculated from the measured R and A_{eff} using the following equations³

$$SE_R = -10 \log(1 - R) \quad (5)$$

$$SE_A = 10 \log(1 - A_{\text{eff}}) \quad (6)$$

$$SE_T = SE_R + SE_A \quad (7)$$

As one can see in Figure 4a, with the addition of graphene fillers up to $\phi \leq 1$ wt%, SE_R does not change and then gradually increases from ~ 0.6 to ~ 3 dB at $\phi = 8$ wt%. In contrast, SE_A increases continuously with graphene loading and reaches ~ 40 dB at $\phi = 8$ wt% at a frequency of $f = 220$ GHz (see Figure 4b). Note that SE_A is constant for the pristine epoxy in the entire frequency range. In contrast, in composites with even a small loading of graphene, SE_A increases as a

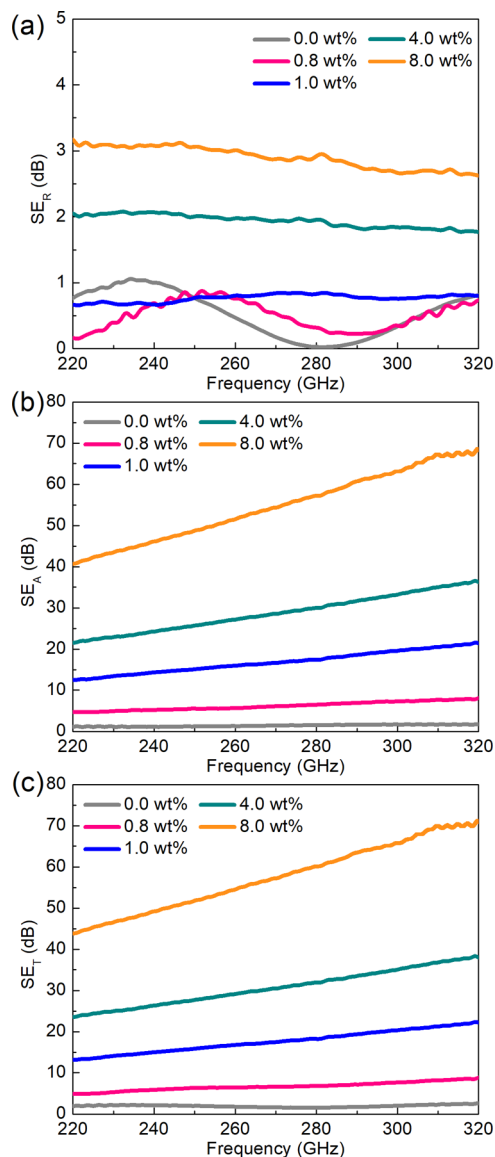


Figure 4. (a) Reflection (SE_R), (b) absorption (SE_A), and (c) total (SE_T) shielding efficiencies of composites at different graphene concentrations. As the filler loading increases, SE_R does not grow significantly whereas SE_A increases substantially. The total shielding efficiency is substantially increased as a result of SE_A enhancement.

function of frequency. The higher the graphene loading into the matrix, the higher is the rate of the increase. For the composite with $\phi = 8$ wt%, SE_A approaches 70 dB at $f = 320$ GHz, which is beyond the requirements of most of the industrial applications. The classical Simon's equation relates SE_A (dB) to frequency, thickness, and electrical resistivity as follows^{3,9,17}

$$SE_A = (1.7t/\rho)f^\gamma \quad (8)$$

where t [cm] is the thickness, ρ [Ωcm] is the bulk resistivity, and f [MHz] is the frequency. Simon's equation assumes a constant frequency exponent $\gamma = 0.5$. Our attempts to fit the experimental data, presented in Figure 4b, with eq 8 and the fixed value of $\gamma = 0.5$ were not successful due to a strong dependence of SE_A on frequency in composites with graphene. We changed the procedure by treating both ρ and γ as the fitting parameters. The latter resulted in obtaining a nearly

constant value of $\gamma = 1.44$ for all concentrations and different values of the effective electrical resistivity for different filler loadings, ϕ . The agreement of the fittings with the experimental data was excellent (see the [Supporting Information](#)). For the composites with graphene, the above equation in the frequency range between $220 \text{ GHz} \leq f \leq 320 \text{ GHz}$ can be revised as

$$SE_A = (1.7t/\rho)f^{1.44} \quad (9)$$

The extracted values of ρ and γ are listed in [Table 1](#). We interpret the extracted ρ as an effective high-frequency resistivity. As described below, the obtained values are reasonable and consistent with DC resistivity measurements.

Table 1. Resistivity and Frequency Exponent for Epoxy with Graphene Fillers

ϕ (wt%)	t [cm]	ρ [$\Omega \text{ cm}$]	γ
0.8	0.1	3.61×10^6	1.495
1.0	0.1	5.65×10^5	1.426
4.0	0.1	2.95×10^5	1.417
8.0	0.1	1.51×10^5	1.415

In [Figure 5](#), we present the absorption shielding efficiency, SE_A , as the function of the graphene filler loading near an

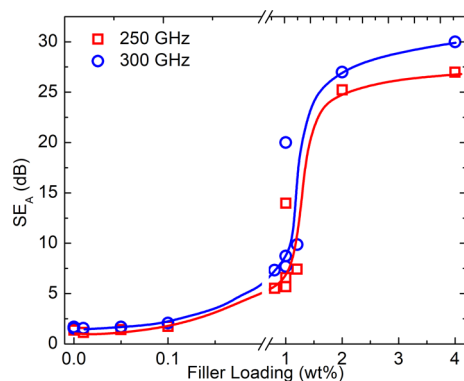


Figure 5. Absorption shielding efficiency as a function of the graphene filler loading. Note an abrupt increase in SE_A at the filler loading, f , in an interval of $1 \text{ wt}\% < f < 2 \text{ wt}\%$.

optimum loading of $\sim 1 \text{ wt}\%$ for the given fillers. The data are presented at two frequencies of 250 and 300 GHz. One can see a strong increase in the shielding efficiency at this loading. At a higher graphene loading, the shielding efficiency increases slower while the reflection grows rapidly. This trend explains the existence of the optimum graphene concentration in the composites for EMI absorbers in this frequency band. We note that there was a substantial SE_A data scatter at the transition point of $\sim 1 \text{ wt}\%$. For this reason, the data points in this region were averaged over results obtained in several measurement runs.

We have measured the DC resistance of the graphene-enhanced composites in the vertical and in-plane, *i.e.*, lateral directions. For the vertical resistance measurements, two high area contacts made of conducting paste were deposited on the top and bottom parts of the sample. The resistance was measured in the two contact configuration. The large cross-sectional area of the contacts allowed us to measure a high resistivity of up to $\sim 10^{13} [\Omega \text{ cm}]$. The in-plane resistance was

measured in the four-probe configuration. For this purpose, four needles contacted the surface of the sample using the probe station. The distance between the contacts was $\sim 0.5 \text{ mm}$. The resistivity was calculated, in a standard way, as $\rho = 2\pi sFR$, where s is the distance between contacts, $F = 0.8$ is the correction factor to take into account for the finite thickness of the sample, and R is the measured resistance.⁷⁶ [Figure 6](#) shows

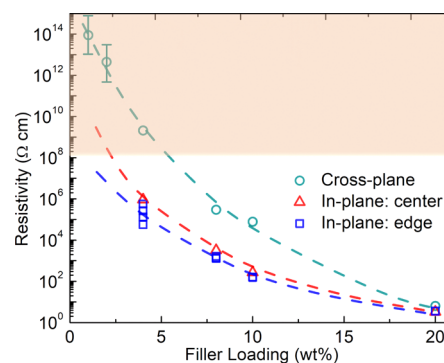


Figure 6. In-plane and cross-plane electrical resistivity of the samples as a function of the graphene loading fraction measured at the DC bias. The in-plane resistivity is smaller than that of the cross-plane direction, indicative that the fillers have preferential in-plane orientation as a result of the sample preparation procedures.

the resistivity of the samples as a function of the graphene filler loading. The shaded area indicates the region, pertinent to the low loading samples, where the resistance was too high to measure in the four-probe configuration. The in-plane resistivity data are shown for the center and peripheral regions of the samples. The in-plane resistivity in the center was consistently slightly smaller than at the edges. One should note that the in-plane resistivity is always smaller than the vertical one. This is an indication that graphene fillers are not completely oriented in random directions and have some preference for the in-plane orientation. The preferential in-plane orientation can provide a positive effect on the EMI shielding properties of the composites. Some degree of the graphene filler alignment was achieved naturally, as a result of the composite preparation procedure, without extra technological steps (see the [Material Synthesis](#)). The measured DC resistivity values for composites with graphene are of the same order of magnitude as those obtained with [eq 9](#). The latter provides an additional validation to our fitting procedure and the formula. The DC resistivity is not expected to be exactly the same as the high-frequency effective resistivity.

The important observation is that the resistivity of composites with the graphene loading below $5 \text{ wt}\%$ is high. The material is electrically insulating, which is beneficial for many EMI shielding applications. The electrical conductivity of the composite is not required for the EM energy absorption and reflection because the EM wave can couple locally to the graphene fillers. We also note that the DC data in [Figure 6](#) do not show an abrupt onset of electrical percolation. There is a rather gradual decrease in the electrical resistivity with increasing graphene loading. From the other side, in [Figure 5](#), we see a drastic increase in SE_A at a loading of $1\text{--}2 \text{ wt}\%$. In [Figure 2a](#), one can also notice a major increase in the reflection, R , when loading changes from 1 to $4 \text{ wt}\%$. We hypothesize that the increase in R and SE_A is related to the onset of electrical percolation at the extremely high-frequency

band. The physics of percolation at high frequency is expected to be different than that at DC or low frequency because of possible contributions of the displacement currents between the graphene fillers separated by narrow dielectric layers. The onset of the electrical percolation at high frequency will lead to the enhanced free-electron type reflection for the whole composite sample. However, more studies are needed to make a definitive conclusion about the mechanism of electrical percolation at high frequency in this type of composites.

The epoxy-based curing composites can serve as adhesives for the electronic components while simultaneously performing the EMI shielding functions. Composites with graphene and other carbon allotropes have been extensively studied for thermal management applications.^{77–88} In this work, we focused on composites with the low graphene loading to provide EMI shielding without EM wave reflection to the environment. At this graphene concentration, the thermal conductivity of the composites is not substantially different from those of the pristine epoxy.⁶⁹ The increase in graphene loading results in the strong enhancement of the heat conduction properties of epoxy-based composites.⁶⁹ The latter results from the excellent thermal conductivity of graphene and FLG.^{89–92} From the other side, a higher loading of graphene fillers also increases the reflection of EM waves as shown in the data in Figures 2 and 3. The strong absorption of the EM wave by the composites means that the energy is transferred from EM waves to heat. In this sense, the higher thermal conductivity of composites can be beneficial. One can envision possible approaches for increasing the thermal conductivity without increasing the EM wave reflection. The first one is the use of layered composites where the inside layer has a higher graphene loading for better heat conduction while the outside layer has a graphene loading of ~1 wt% for the optimum EMI shielding without reflection. The second approach is to use a combination of graphene and boron nitride (BN) fillers with a larger total loading. The BN fillers, unlike graphene fillers, are not electrically conductive but also have much higher thermal conductivity than the epoxy base.^{69,72,93} As graphene is added to pure epoxy, one would expect an increase in Young's modulus and decrease in the ultimate tensile strength (UTS). However, the changes are not substantial since the amount of graphene loaded to the matrix is low. More importantly, the fracture toughness of the pure epoxy is improved at low graphene loadings since the fillers can efficiently deflect the propagating cracks in the base epoxy matrix.^{94–96} The extra benefits of a low graphene filler loading is that the composites are cost-effective and can be easily molded and processed. The composites and coating layers with graphene fillers and related carbon allotropes have also demonstrated reliability and robustness against thermal cycling.^{70,97–100}

IV. CONCLUSIONS

We report on the synthesis of the epoxy-based composites with graphene fillers and test their EMI shielding efficiency in the sub-terahertz frequency range (band: 220–320 GHz). It was found that the electromagnetic transmission, T , is low even at small concentrations of graphene fillers: $T < 1\%$ at a frequency of 300 GHz for a composite with 1wt% graphene fillers. Our results demonstrate that graphene epoxy-based composites provide efficient electromagnetic shielding in the sub-terahertz band via absorption and small energy reflection to the environment. At a frequency of 320 GHz, we achieved a maximum total shielding efficiency of ~70 dB for epoxy with

only 8 wt% graphene loading. This EMI shielding is beyond the requirements of most of the industrial applications. The developed lightweight composites with graphene can be used as electromagnetic absorbers in the microwave radio relays, remote sensors, millimeter wave scanners, and wireless local area networks. Since our composites are curing materials, they can be used as adhesives for packaging the electronic components while simultaneously performing the EMI shielding functions.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsami.0c06729>.

SEM character of the FLG and epoxy with 1 wt % FLG loading and linear fitting of absorption shielding efficiency using Simon's equation (PDF)

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Author Contributions

A.A.B. and S.R. conceived the idea of the study, coordinated the project, and contributed to the EM data analysis. K.G. and Y.Y. performed the EM shielding measurements. Z.B. prepared the composites, performed material characterization, and assisted with the EM data analysis. F.K. contributed to the sample preparation and EM data analysis. A.R. performed DC resistivity measurements. G.C. and W.K. contributed to data analysis. A.A.B. and S.R. led the manuscript preparation. All authors contributed to writing and editing of the manuscript.

Notes

The authors declare no competing financial interest.

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REFERENCES

- (1) Jiang, D.; Murugadoss, V.; Wang, Y.; Lin, J.; Ding, T.; Wang, Z.; Shao, Q.; Wang, C.; Liu, H.; Lu, N.; Wei, R.; Subramania, A.; Guo, Z. Electromagnetic Interference Shielding Polymers and Nanocomposites - A Review. *Polym. Rev* **2019**, *59*, 280–337.
- (2) Wilson, R.; George, G.; Joseph, K. *An Introduction to Materials for Potential EMI Shielding Applications: Status and Future*; Joseph, K., Wilson, R.; George, G.; Eds.; Elsevier Inc., 2020.
- (3) Kargar, F.; Barani, Z.; Balinskiy, M.; Magana, A. S.; Lewis, J. S.; Balandin, A. A. Dual-Functional Graphene Composites for Electromagnetic Shielding and Thermal Management. *Adv. Electron. Mater.* **2019**, *5*, 1800558.
- (4) Deruelle, F. The Different Sources of Electromagnetic Fields: Dangers Are Not Limited to Physical Health. *Electromagn. Biol. Med.* **2020**, *39*, 166–175.
- (5) Kostoff, R. N.; Heroux, P.; Aschner, M.; Tsatsakis, A. Adverse Health Effects of 5G Mobile Networking Technology under Real-Life Conditions. *Toxicol. Lett.* **2020**, *323*, 35–40.
- (6) Repacholi, M. H. Low-Level Exposure to Radiofrequency Electromagnetic Fields: Health Effects and Research Needs. *Bioelectromagnetics* **1998**, *19*, 1–19.
- (7) Zamanian, A.; Hardiman, C. Electromagnetic Radiation and Human Health: A Review of Sources and Effects. *EMR Hum. Heal.* **2005**, *16*, 16–26.
- (8) Hardell, L.; Sage, C. Biological Effects from Electromagnetic Field Exposure and Public Exposure Standards. *Biomed. Pharmacother.* **2008**, *62*, 104–109.
- (9) Simon, R. M. EMI Shielding Through Conductive Plastics. *Polym.-Plast. Technol. Eng.* **1981**, *17*, 1–10.
- (10) Bigg, D. M.; Stutz, D. E. Plastic Composites for Electromagnetic Interference Shielding Applications. *Polym. Compos.* **1983**, *4*, 40–46.
- (11) Li, L.; Chung, D. D. L. Electrical and Mechanical Properties of Electrically Conductive Polyethersulfone Composites. *Composites* **1994**, *25*, 215–224.
- (12) Lu, G.; Li, X.; Jiang, H. Electrical and Shielding Properties of ABS Resin Filled with Nickel-Coated Carbon Fibers. *Compos. Sci. Technol.* **1996**, *56*, 193–200.
- (13) Chung, D. D. L. Materials for Electromagnetic Interference Shielding. *J. Mater. Eng. Perform.* **2000**, *9*, 350–354.
- (14) Chung, D. D. L. Electromagnetic Interference Shielding Effectiveness of Carbon Materials. *Carbon* **2001**, *39*, 279–285.
- (15) Wan, Y.-J.; Li, G.; Yao, Y.-M.; Zeng, X.-L.; Zhu, P.-L.; Sun, R. Recent Advances in Polymer-Based Electronic Packaging Materials. *Compos. Commun.* **2020**, *19*, 154–167.
- (16) Wanasinghe, D.; Aslani, F.; Ma, G.; Habibi, D. Review of Polymer Composites with Diverse Nanofillers for Electromagnetic Interference Shielding. *Nanomaterials* **2020**, *10*, 541.
- (17) Shahzad, F.; Alhabeib, M.; Hatter, C. B.; Anasori, B.; Man Hong, S.; Koo, C. M.; Gogotsi, Y. Electromagnetic Interference Shielding with 2D Transition Metal Carbides (MXenes). *Science* **2016**, *353*, 1137–1140.
- (18) Roh, J.-S.; Chi, Y.-S.; Kang, T. J.; Nam, S.-w. Electromagnetic Shielding Effectiveness of Multifunctional Metal Composite Fabrics. *Text. Res. J.* **2008**, *78*, 825–835.
- (19) Wenderoth, K.; Petermann, J.; Kruse, K.-D.; ter Haseborg, J.-L.; Krieger, W. Synergism on Electromagnetic Inductance (EMI)-shielding in Metal- and Ferroelectric-particle Filled Polymers. *Polym. Compos.* **1989**, *10*, 52–56.
- (20) Song, W.-L.; Wang, J.; Fan, L.-Z.; Li, Y.; Wang, C.-Y.; Cao, M.-S. Interfacial Engineering of Carbon Nanofiber–Graphene–Carbon Nanofiber Heterojunctions in Flexible Lightweight Electromagnetic Shielding Networks. *ACS Appl. Mater. Interfaces* **2014**, *6*, 10516–10523.
- (21) Yang, Y.; Gupta, M. C.; Dudley, K. L. Towards Cost-Efficient EMI Shielding Materials Using Carbon Nanostructure-Based Nanocomposites. *Nanotechnology* **2007**, *18*, 345701.
- (22) Yang, Y.; Gupta, M. C.; Dudley, K. L.; Lawrence, R. W. Conductive Carbon Nanofiber-Polymer Foam Structures. *Adv. Mater.* **2005**, *17*, 1999–2003.
- (23) Ameli, A.; Jung, P. U.; Park, C. B. Electrical Properties and Electromagnetic Interference Shielding Effectiveness of Polypropylene/Carbon Fiber Composite Foams. *Carbon* **2013**, *60*, 379–391.
- (24) Lee, B. O.; Woo, W. J.; Park, H. S.; Hahm, H. S.; Wu, J. P.; Kim, M. S. Influence of Aspect Ratio and Skin Effect on EMI Shielding of Coating Materials Fabricated with Carbon Nanofiber/PVDF. *J. Mater. Sci.* **2002**, *37*, 1839–1843.
- (25) Bayat, M.; Yang, H.; Ko, F. K.; Michelson, D.; Mei, A. Electromagnetic Interference Shielding Effectiveness of Hybrid Multifunctional Fe₃O₄/Carbon Nanofiber Composite. *Polymer* **2014**, *55*, 936–943.
- (26) Crespo, M.; Méndez, N.; González, M.; Baselga, J.; Pozuelo, J. Synergistic Effect of Magnetite Nanoparticles and Carbon Nanofibres in Electromagnetic Absorbing Composites. *Carbon* **2014**, *74*, 63–72.
- (27) Wu, J.; Ye, Z.; Ge, H.; Chen, J.; Liu, W.; Liu, Z. Modified Carbon Fiber/Magnetic Graphene/Epoxy Composites with Synergistic Effect for Electromagnetic Interference Shielding over Broad Frequency Band. *J. Colloid Interface Sci.* **2017**, *506*, 217–226.
- (28) Mondal, S.; Ganguly, S.; Das, P.; Khashtgir, D.; Das, N. C. Low Percolation Threshold and Electromagnetic Shielding Effectiveness of Nano-Structured Carbon Based Ethylene Methyl Acrylate Nanocomposites. *Compos. Part B Eng.* **2017**, *119*, 41–56.
- (29) Strakhov, I. S.; Rodnaya, A. I.; Mezhev, Y. O.; Korshak, Y. V.; Vagramyan, T. A. Enhancement of the Strength of a Composite Material Based on ED-20 Epoxy Resin by Reinforcement with a Carbon Fiber Modified by Electrochemical Deposition of Poly(o-Phenylenediamine). *Russ. J. Appl. Chem.* **2014**, *87*, 1918–1922.
- (30) Kuester, S.; Merlini, C.; Barra, G. M. O.; Ferreira, J. C., Jr.; Lucas, A.; de Souza, A. C.; Soares, B. G. Processing and Characterization of Conductive Composites Based on Poly(Styrene-*b*-Ethylene-Ran-Butylene-*b*-Styrene) (SEBS) and Carbon Additives: A Comparative Study of Expanded Graphite and Carbon Black. *Compos. Part B Eng.* **2016**, *84*, 236–247.

- (31) De Bellis, G.; Tamburrano, A.; Dinescu, A.; Santarelli, M. L.; Sarto, M. S. Electromagnetic Properties of Composites Containing Graphite Nanoplatelets at Radio Frequency. *Carbon* **2011**, *49*, 4291–4300.
- (32) Panwar, V.; Mehra, R. M. Analysis of Electrical, Dielectric, and Electromagnetic Interference Shielding Behavior of Graphite Filled High Density Polyethylene Composites. *Polym. Eng. Sci.* **2008**, *48*, 2178–2187.
- (33) Jiang, X.; Yan, D.-X.; Bao, Y.; Pang, H.; Ji, X.; Li, Z.-M. Facile, Green and Affordable Strategy for Structuring Natural Graphite/Polymer Composite with Efficient Electromagnetic Interference Shielding. *RSC Adv.* **2015**, *5*, 22587–22592.
- (34) Al-Saleh, M. H.; Sundararaj, U. Electromagnetic Interference Shielding Mechanisms of CNT/Polymer Composites. *Carbon* **2009**, *47*, 1738–1746.
- (35) Yang, Y.; Gupta, M. C.; Dudley, K. L.; Lawrence, R. W. Novel Carbon Nanotube - Polystyrene Foam Composites for Electromagnetic Interference Shielding. *Nano Lett.* **2005**, *5*, 2131–2134.
- (36) Das, N. C.; Liu, Y.; Yang, K.; Peng, W.; Maiti, S.; Wang, H. Single-Walled Carbon Nanotube/Poly(Methyl Methacrylate) Composites for Electromagnetic Interference Shielding. *Polym. Eng. Sci.* **2009**, *49*, 1627–1634.
- (37) Kausar, A.; Ahmad, S.; Salman, S. M. Effectiveness of Polystyrene/Carbon Nanotube Composite in Electromagnetic Interference Shielding Materials: A Review. *Polym. - Plast. Technol. Eng.* **2016**, *56*, 1027–1042.
- (38) Thomassin, J.-M.; Jérôme, C.; Pardoën, T.; Bailly, C.; Huynen, I.; Detrembleur, C. Polymer/Carbon Based Composites as Electromagnetic Interference (EMI) Shielding Materials. *Mater. Sci. Eng. R* **2013**, *74*, 211–232.
- (39) Lin, J.-H.; Lin, Z.-I.; Pan, Y.-J.; Huang, C.-L.; Chen, C.-K.; Lou, C.-W. Polymer Composites Made of Multi-Walled Carbon Nanotubes and Graphene Nano-Sheets: Effects of Sandwich Structures on Their Electromagnetic Interference Shielding Effectiveness. *Compos. Part B Eng.* **2016**, *89*, 424–431.
- (40) Chung, D. D. L. Carbon Materials for Structural Self-Sensing, Electromagnetic Shielding and Thermal Interfacing. *Carbon* **2012**, *50*, 3342–3353.
- (41) Ling, J.; Zhai, W.; Feng, W.; Shen, B.; Zhang, J.; Zheng, W. g. Facile Preparation of Lightweight Microcellular Polyetherimide/Graphene Composite Foams for Electromagnetic Interference Shielding. *ACS Appl. Mater. Interfaces* **2013**, *5*, 2677–2684.
- (42) Hsiao, S.-T.; Ma, C.-C. M.; Liao, W.-H.; Wang, Y.-S.; Li, S.-M.; Huang, Y.-C.; Yang, R.-B.; Liang, W.-F. Lightweight and Flexible Reduced Graphene Oxide/Water-Borne Polyurethane Composites with High Electrical Conductivity and Excellent Electromagnetic Interference Shielding Performance. *ACS Appl. Mater. Interfaces* **2014**, *6*, 10667–10678.
- (43) Mazzoli, A.; Corinaldesi, V.; Donnini, J.; Di Perna, C.; Micheli, D.; Vricella, A.; Pastore, R.; Bastianelli, L.; Moglie, F.; Mariani Primiani, V. Effect of Graphene Oxide and Metallic Fibers on the Electromagnetic Shielding Effect of Engineered Cementitious Composites. *J. Build. Eng.* **2018**, *18*, 33–39.
- (44) Yang, W.; Zhao, Z.; Wu, K.; Huang, R.; Liu, T.; Jiang, H.; Chen, F.; Fu, Q. Ultrathin Flexible Reduced Graphene Oxide/Cellulose Nanofiber Composite Films with Strongly Anisotropic Thermal Conductivity and Efficient Electromagnetic Interference Shielding. *J. Mater. Chem. C* **2017**, *5*, 3748–3756.
- (45) Sharif, F.; Arjmand, M.; Moud, A. A.; Sundararaj, U.; Roberts, E. P. L. Segregated Hybrid Poly(Methyl Methacrylate)/Graphene/Magnetite Nanocomposites for Electromagnetic Interference Shielding. *ACS Appl. Mater. Interfaces* **2017**, *9*, 14171–14179.
- (46) Bagotia, N.; Choudhary, V.; Sharma, D. K. Superior Electrical, Mechanical and Electromagnetic Interference Shielding Properties of Polycarbonate/Ethylene-Methyl Acrylate-in Situ Reduced Graphene Oxide Nanocomposites. *J. Mater. Sci.* **2018**, *53*, 16047–16061.
- (47) Yan, D.-X.; Ren, P.-G.; Pang, H.; Fu, Q.; Yang, M.-B.; Li, Z.-M. Efficient Electromagnetic Interference Shielding of Lightweight Graphene/Polystyrene Composite. *J. Mater. Chem.* **2012**, *22*, 18772.
- (48) Yuan, B.; Yu, L.; Sheng, L.; An, K.; Zhao, X. Comparison of Electromagnetic Interference Shielding Properties between Single-Wall Carbon Nanotube and Graphene Sheet/Polyaniline Composites. *J. Phys. D: Appl. Phys.* **2012**, *45*, 235108.
- (49) Yuan, B.; Bao, C.; Qian, X.; Song, L.; Tai, Q.; Liew, K. M.; Hu, Y. Design of Artificial Nacre-like Hybrid Films as Shielding to Mitigate Electromagnetic Pollution. *Carbon* **2014**, *75*, 178–189.
- (50) Wen, B.; Wang, X. X.; Cao, W. Q.; Shi, H. L.; Lu, M. M.; Wang, G.; Jin, H. B.; Wang, W. Z.; Yuan, J.; Cao, M. S. Reduced Graphene Oxides: The Thinnest and Most Lightweight Materials with Highly Efficient Microwave Attenuation Performances of the Carbon World. *Nanoscale* **2014**, *6*, 5754–5761.
- (51) Chen, Z.; Xu, C.; Ma, C.; Ren, W.; Cheng, H.-M. Lightweight and Flexible Graphene Foam Composites for High-Performance Electromagnetic Interference Shielding. *Adv. Mater.* **2013**, *25*, 1296–1300.
- (52) Yan, D.-X.; Pang, H.; Li, B.; Vajtai, R.; Xu, L.; Ren, P.-G.; Wang, J.-H.; Li, Z.-M. Structured Reduced Graphene Oxide/Polymer Composites for Ultra-Efficient Electromagnetic Interference Shielding. *Adv. Funct. Mater.* **2015**, *25*, 559–566.
- (53) Liang, J.; Wang, Y.; Huang, Y.; Ma, Y.; Liu, Z.; Cai, J.; Zhang, C.; Gao, H.; Chen, Y. Electromagnetic Interference Shielding of Graphene/Epoxy Composites. *Carbon* **2009**, *47*, 922–925.
- (54) Dalal, J.; Lather, S.; Gupta, A.; Dahiya, S.; Maan, A. S.; Singh, K.; Dhawan, S. K.; Ohlan, A. EMI Shielding Properties of Laminated Graphene and PbTiO₃ Reinforced Poly(3,4-Ethylenedioxythiophene) Nanocomposites. *Compos. Sci. Technol.* **2018**, *165*, 222–230.
- (55) Ren, F.; Song, D.; Li, Z.; Jia, L.; Zhao, Y.; Yan, D.; Ren, P. Synergistic Effect of Graphene Nanosheets and Carbonyl Iron-Nickel Alloy Hybrid Filler on Electromagnetic Interference Shielding and Thermal Conductivity of Cyanate Ester Composites. *J. Mater. Chem. C* **2018**, *6*, 1476–1486.
- (56) Shen, B.; Li, Y.; Yi, D.; Zhai, W.; Wei, X.; Zheng, W. Strong Flexible Polymer/Graphene Composite Films with 3D Saw-Tooth Folding for Enhanced and Tunable Electromagnetic Shielding. *Carbon* **2017**, *113*, 55–62.
- (57) Zhao, B.; Zhao, C.; Li, R.; Hamidinejad, S. M.; Park, C. B. Flexible, Ultrathin, and High-Efficiency Electromagnetic Shielding Properties of Poly(Vinylidene Fluoride)/Carbon Composite Films. *ACS Appl. Mater. Interfaces* **2017**, *9*, 20873–20884.
- (58) Li, Y.; Shen, B.; Yi, D.; Zhang, L.; Zhai, W.; Wei, X.; Zheng, W. The Influence of Gradient and Sandwich Configurations on the Electromagnetic Interference Shielding Performance of Multilayered Thermoplastic Polyurethane/Graphene Composite Foams. *Compos. Sci. Technol.* **2017**, *138*, 209–216.
- (59) Song, Q.; Ye, F.; Yin, X.; Li, W.; Li, H.; Liu, Y.; Li, K.; Xie, K.; Li, X.; Fu, Q.; Cheng, L.; Zhang, L.; Wei, B. Carbon Nanotube–Multilayered Graphene Edge Plane Core–Shell Hybrid Foams for Ultrahigh-Performance Electromagnetic-Interference Shielding. *Adv. Mater.* **2017**, *29*, 1701583.
- (60) Bagotia, N.; Choudhary, V.; Sharma, D. K. Synergistic Effect of Graphene/Multiwalled Carbon Nanotube Hybrid Fillers on Mechanical, Electrical and EMI Shielding Properties of Polycarbonate/Ethylene Methyl Acrylate Nanocomposites. *Compos. Part B Eng.* **2019**, *159*, 378–388.
- (61) Zhao, B.; Wang, S.; Zhao, C.; Li, R.; Hamidinejad, S. M.; Kazemi, Y.; Park, C. B. Synergism between Carbon Materials and Ni Chains in Flexible Poly(Vinylidene Fluoride) Composite Films with High Heat Dissipation to Improve Electromagnetic Shielding Properties. *Carbon* **2018**, *127*, 469–478.
- (62) Zhan, Y.; Wang, J.; Zhang, K.; Li, Y.; Meng, Y.; Yan, N.; Wei, W.; Peng, F.; Xia, H. Fabrication of a Flexible Electromagnetic Interference Shielding Fe₃O₄@reduced Graphene Oxide/Natural Rubber Composite with Segregated Network. *Chem. Eng. J.* **2018**, *344*, 184–193.
- (63) Min, Z.; Yang, H.; Chen, F.; Kuang, T. Scale-up Production of Lightweight High-Strength Polystyrene/Carbonaceous Filler Composite Foams with High-Performance Electromagnetic Interference Shielding. *Mater. Lett.* **2018**, *230*, 157–160.

- (64) Lee, S.-H.; Kang, D.; Oh, I.-K. Multilayered Graphene-Carbon Nanotube-Iron Oxide Three-Dimensional Heterostructure for Flexible Electromagnetic Interference Shielding Film. *Carbon* **2017**, *111*, 248–257.
- (65) Verma, M.; Chauhan, S. S.; Dhawan, S. K.; Choudhary, V. Graphene Nanoplatelets/Carbon Nanotubes/Polyurethane Composites as Efficient Shield against Electromagnetic Polluting Radiations. *Compos. Part B Eng.* **2017**, *120*, 118–127.
- (66) Yang, L.; Chen, Y.; Wang, M.; Shi, S.; Jing, J. Fused Deposition Modeling 3D Printing of Novel Poly (Vinyl Alcohol)/Graphene Nanocomposite with Enhanced Mechanical and Electromagnetic Interference Shielding Properties. *Ind. Eng. Chem. Res.* **2020**, 8066.
- (67) Lan, C.; Zou, L.; Qiu, Y.; Ma, Y. Tuning Solid–Air Interface of Porous Graphene Paper for Enhanced Electromagnetic Interference Shielding. *J. Mater. Sci.* **2020**, *55*, 6598–6609.
- (68) Zdrojek, M.; Bomba, J.; Lapińska, A.; Dużyńska, A.; Żerańska-Chudek, K.; Suszek, J.; Stobiński, L.; Taube, A.; Sypek, M.; Judek, J. Graphene-Based Plastic Absorber for Total Sub-Terahertz Radiation Shielding. *Nanoscale* **2018**, *10*, 13426–13431.
- (69) Kargar, F.; Barani, Z.; Salgado, R.; Debnath, B.; Lewis, J. S.; Aytan, E.; Lake, R. K.; Balandin, A. A. Thermal Percolation Threshold and Thermal Properties of Composites with High Loading of Graphene and Boron Nitride Fillers. *ACS Appl. Mater. Interfaces* **2018**, *10*, 37555–37565.
- (70) Naghibi, S.; Kargar, F.; Wright, D.; Huang, C. Y. T.; Mohammadzadeh, A.; Barani, Z.; Salgado, R.; Balandin, A. A. Noncuring Graphene Thermal Interface Materials for Advanced Electronics. *Adv. Electron. Mater.* **2020**, 1901303.
- (71) Barani, Z.; Mohammadzadeh, A.; Geremew, A.; Huang, C.-Y.; Coleman, D.; Mangolini, L.; Kargar, F.; Balandin, A. A. Thermal Properties of the Binary-Filler Hybrid Composites with Graphene and Copper Nanoparticles. *Adv. Funct. Mater.* **2019**, 1904008.
- (72) Lewis, J. S.; Barani, Z.; Magana, A. S.; Kargar, F.; Balandin, A. A. Thermal and Electrical Conductivity Control in Hybrid Composites with Graphene and Boron Nitride Fillers. *Mater. Res. Express* **2019**, *6*, No. 085325.
- (73) Dvurechenskaya, N.; Bajurko, P. R.; Zieliński, R. J.; Yashchishyn, Y. Measurements of Shielding Effectiveness of Textile Materials Containing Metal by the Free-Space Transmission Technique with Data Processing in the Time Domain. *Metrol. Meas. Syst.* **2013**, *21*, 217–228.
- (74) Yashchishyn, Y.; Godziszewski, K. A New Method for Dielectric Characterization in Sub-THz Frequency Range. *IEEE Trans. Terahertz Sci. Technol.* **2018**, *8*, 19–26.
- (75) Godziszewski, K.; Yashchishyn, Y. Investigation of Influence of Measurement Conditions on Accuracy of Material Characterization in Sub-THz Frequency Range. In *2016 21st International Conference on Microwave, Radar and Wireless Communications, MIKON 2016*; Institute of Electrical and Electronics Engineers Inc., 2016.
- (76) Schroder, D. K. *Semiconductor Material and Device Characterization*; John Wiley & Sons, 2015.
- (77) Ohayon-Lavi, A.; Buzaglo, M.; Ligati, S.; Peretz-Damari, S.; Shachar, G.; Pinsk, N.; Riskin, M.; Schatzberg, Y.; Genish, I.; Regev, O. Compression-Enhanced Thermal Conductivity of Carbon Loaded Polymer Composites. *Carbon* **2020**, *163*, 333–340.
- (78) Shtein, M.; Nadiv, R.; Buzaglo, M.; Regev, O. Graphene-Based Hybrid Composites for Efficient Thermal Management of Electronic Devices. *ACS Appl. Mater. Interfaces* **2015**, *7*, 23725–23730.
- (79) Yuan, W.; Xiao, Q.; Li, L.; Xu, T. Thermal Conductivity of Epoxy Adhesive Enhanced by Hybrid Graphene Oxide/AlN Particles. *Appl. Therm. Eng.* **2016**, *106*, 1067–1074.
- (80) Yue, L.; Pircheraghi, G.; Monemian, S. A.; Manas-Zloczower, I. Epoxy Composites with Carbon Nanotubes and Graphene Nanoplatelets – Dispersion and Synergy Effects. *Carbon* **2014**, *78*, 268–278.
- (81) Shtein, M.; Nadiv, R.; Buzaglo, M.; Kahil, K.; Regev, O. Thermally Conductive Graphene-Polymer Composites: Size, Percolation, and Synergy Effects. *Chem. Mater.* **2015**, *27*, 2100–2106.
- (82) Liang, X.; Dai, F. Epoxy Nanocomposites with Reduced Graphene Oxide-Constructed Three-Dimensional Networks of Single Wall Carbon Nanotube for Enhanced Thermal Management Capability with Low Filler Loading. *ACS Appl. Mater. Interfaces* **2019**, *12*, 3051–3058.
- (83) Feng, Y.; Li, X.; Zhao, X.; Ye, Y.; Zhou, X.; Liu, H.; Liu, C.; Xie, X. Synergetic Improvement in Thermal Conductivity and Flame Retardancy of Epoxy/Silver Nanowires Composites by Incorporating “Branch-Like” Flame-Retardant Functionalized Graphene. *ACS Appl. Mater. Interfaces* **2018**, *10*, 21628–21641.
- (84) Guo, H.; Li, X.; Li, B.; Wang, J.; Wang, S. Thermal Conductivity of Graphene/Poly(Vinylidene Fluoride) Nanocomposite Membrane. *Mater. Des.* **2017**, *114*, 355–363.
- (85) Bento, J. L.; Brown, E.; Woltornist, S. J.; Adamson, D. H. Thermal and Electrical Properties of Nanocomposites Based on Self-Assembled Pristine Graphene. *Adv. Funct. Mater.* **2017**, *27*, 1604277.
- (86) Li, A.; Zhang, C.; Zhang, Y.-F. Thermal Conductivity of Graphene-Polymer Composites: Mechanisms, Properties, and Applications. *Polymer* **2017**, *9*, 437.
- (87) Shao, L.; Shi, L.; Li, X.; Song, N.; Ding, P. Synergistic Effect of BN and Graphene Nanosheets in 3D Framework on the Enhancement of Thermal Conductive Properties of Polymeric Composites. *Compos. Sci. Technol.* **2016**, *135*, 83–91.
- (88) Pathak, A. K.; Borah, M.; Gupta, A.; Yokozeki, T.; Dhakate, S. R. Improved Mechanical Properties of Carbon Fiber/Graphene Oxide-Epoxy Hybrid Composites. *Compos. Sci. Technol.* **2016**, *135*, 28–38.
- (89) Balandin, A. A.; Ghosh, S.; Bao, W.; Calizo, I.; Teweldebrhan, D.; Miao, F.; Lau, C. N. Superior Thermal Conductivity of Single-Layer Graphene. *Nano Lett.* **2008**, *8*, 902–907.
- (90) Fugallo, G.; Cepellotti, A.; Paulatto, L.; Lazzeri, M.; Marzari, N.; Mauri, F. Thermal Conductivity of Graphene and Graphite: Collective Excitations and Mean Free Paths. *Nano Lett.* **2014**, *14*, 6109–6114.
- (91) Shahil, K. M. F.; Balandin, A. A. Graphene-Multilayer Graphene Nanocomposites as Highly Efficient Thermal Interface Materials. *Nano Lett.* **2012**, *12*, 861–867.
- (92) Nika, D. L.; Ghosh, S.; Pokatilov, E. P.; Balandin, A. A. Lattice Thermal Conductivity of Graphene Flakes: Comparison with Bulk Graphite. *Appl. Phys. Lett.* **2009**, *94*, 203103.
- (93) Kargar, F.; Salgado, R.; Legedza, S.; Renteria, J.; Balandin, A. A. A Comparative Study of the Thermal Interface Materials with Graphene and Boron Nitride Fillers. In *Carbon Nanotubes, Graphene, and Associated Devices VII*; International Society for Optics and Photonics, 2014; Vol. 9168, pp 91680S–91680S – 5.
- (94) Atif, R.; Shyha, I.; Inam, F. Mechanical, Thermal, and Electrical Properties of Graphene-Epoxy Nanocomposites-A Review. *Polymer* **2016**, *8*, 281.
- (95) Naebe, M.; Wang, J.; Amini, A.; Khayyam, H.; Hameed, N.; Li, L. H.; Chen, Y.; Fox, B. Mechanical Property and Structure of Covalent Functionalised Graphene/Epoxy Nanocomposites. *Sci. Rep.* **2015**, *4*, 4375.
- (96) King, J. A.; Klimek, D. R.; Miskioğlu, I.; Odegard, G. M. Mechanical Properties of Graphene Nanoplatelet/Epoxy Composites. *J. Compos. Mater.* **2014**, *49*, 659–668.
- (97) Lewis, J. S.; Perrier, T.; Mohammadzadeh, A.; Kargar, F.; Balandin, A. A. Power Cycling and Reliability Testing of Epoxy-Based Graphene Thermal Interface Materials. *C — J. Carbon Res.* **2020**, *6*, 26.
- (98) Mahadevan, B. K.; Naghibi, S.; Kargar, F.; Balandin, A. A. Non-Curing Thermal Interface Materials with Graphene Fillers for Thermal Management of Concentrated Photovoltaic Solar Cells. *C — J. Carbon Res.* **2020**, *6*, 2.
- (99) Renteria, J. D.; Ramirez, S.; Malekpour, H.; Alonso, B.; Centeno, A.; Zurutuza, A.; Cocemasov, A. I.; Nika, D. L.; Balandin, A. A. Strongly Anisotropic Thermal Conductivity of Free-Standing Reduced Graphene Oxide Films Annealed at High Temperature. *Adv. Funct. Mater.* **2015**, *25*, 4664–4672.

(100) Balandin, A. A. Phononics of Graphene and Related Materials.
ACS Nano **2020**, 5170.